

This Page Is Inserted by IFW Operations
and is not a part of the Official Record

BEST AVAILABLE IMAGES

Defective images within this document are accurate representations of the original documents submitted by the applicant.

Defects in the images may include (but are not limited to):

- BLACK BORDERS
- TEXT CUT OFF AT TOP, BOTTOM OR SIDES
- FADED TEXT
- ILLEGIBLE TEXT
- SKEWED/SLANTED IMAGES
- COLORED PHOTOS
- BLACK OR VERY BLACK AND WHITE DARK PHOTOS
- GRAY SCALE DOCUMENTS

IMAGES ARE BEST AVAILABLE COPY.

**As rescanning documents *will not* correct images,
please do not report the images to the
Image Problem Mailbox.**

(19)



Europäisches Patentamt
European Patent Office
Office européen des brevets

(11) Publication number:

0 142 931
A1

(12)

EUROPEAN PATENT APPLICATION

(21) Application number: 84306854.5

(51) Int. Cl.⁴: C 09 K 11/77

(22) Date of filing: 09.10.84

(30) Priority: 17.10.83 US 542391

(43) Date of publication of application:
29.05.85 Bulletin 85/22(84) Designated Contracting States:
DE FR GB NL SE

(71) Applicant: AMERICAN TELEPHONE AND TELEGRAPH
COMPANY
550 Madison Avenue
New York, NY 10022(US)

(72) Inventor: Berkstresser, George Wayne
1022 Papen Road
Bridgewater New Jersey 08807(US)

(72) Inventor: Huo, Tai-Chan Donald
777 Mountain Avenue
New Providence New Jersey 07974(US)

(72) Inventor: Shmulovich, Joseph
33 Westview Avenue
New Providence New Jersey 07974(US)

(74) Representative: Johnston, Kenneth Graham et al,
Western Electric Company Limited 5 Mornington Road
Woodford Green Essex, IG8 OTU(GB)

(54) Visual display system utilizing high luminosity single crystal garnet material.

(57) This invention is a display apparatus comprising a novel high intensity broadband-emitting phosphor. The phosphor is single crystal material in which Tb acts as sensitizer of Ce, the activator. The composition of the phosphor is $(Y_{3-x-y-z}Ce_xTb_yRE_z)Al_{5-w}X_wO_{12}$, where RE is one or more 4f-type rare earths other than Y, Ce, and Tb, X is Sc, Ga, or In, and $0 < x \leq 0.06$, $0 < y \leq 0.5$, $0 \leq z \leq 2$, $0 \leq w \leq 3$. In preferred embodiments z and w are either zero, or RE is Lu and X is Ga. The phosphor is typically grown epitaxially by LPE on a $Y_3Al_5O_{12}$ substrate, and has application in a variety of display apparatus, including aircraft cockpit head-up displays and flight simulators.

EP 0 142 931 A1

VISUAL DISPLAY SYSTEM UTILIZING
HIGH LUMINOSITY SINGLE CRYSTAL GARNET MATERIAL

Field of the Invention

5 This invention pertains to the field of
visual display systems utilizing inorganic phosphors.

Background of the Invention

 This invention concerns visual display devices
utilizing inorganic phosphors that emit visible
10 electromagnetic radiation when excited by energetic
electrons (cathodoluminescence, CL) or by electromagnetic
radiation (photoluminescence, PL), and the discussion
herein will be so limited.

 Although visual display technology utilizing
15 phosphors, especially cathode ray tube (CRT) technology, is
the product of many years of intense effort and is quite
mature, there exist numerous potential applications for
such displays that are currently unrealized or only
imperfectly realized. These applications share a common
20 requirement of high phosphor luminosity. Among these
applications are projection displays, high resolution
displays, and high brightness displays for viewing in the
presence of high ambient light levels, e.g., avionic
displays, such as head-up displays and flight simulators.
25 Although novel technologies are being developed for these
and other applications, e.g., holographic combiners, that
require intense narrow-band luminescence emission, the
existence of a well-developed CRT display technology based
on broadband emission suggests that development of high
30 intensity broadband phosphors might be an economically
advantageous approach to the realization of such devices.

 Prior art display devices often employ powder
phosphors. These phosphors are typically subject to
shortcomings such as poor thermal properties, degradation
35 of efficiency with dose, and limited resolution. Many of
these shortcomings of prior art powder phosphor targets can
be alleviated by the use of single crystal phosphor

targets, typically consisting of a monolithic slab of crystalline phosphor that is bonded at the atomic level to a single crystal faceplate. Such epitaxial targets are typically grown by liquid phase epitaxy (LPE). See, for instance, J. M. Robertson et al, Philips Journal of Research, Vol. 35, pp. 354-371, (1980). Single crystal targets have been found to be capable of withstanding input power densities in excess of 10^{10} W/m² without showing measurable degradation, and can have improved resolution as compared to powder phosphor targets.

The principles of luminescence in inorganic solids are well known and will not be reviewed herein in detail. (See, for instance, Luminescence of Inorganic Solids, P. Goldberg, Editor, Academic Press, New York (1966), especially pp. 2-52.) Briefly, luminescent emission from inorganic solids involves optical transitions between electronic states characteristic of the radiating substance. The radiating entity, e.g., an atom occupying a crystal lattice site, is raised into an excited state through interaction with the excitation means, e.g., a UV photon or an energetic electron, followed by the entity's return to the electronic ground state, typically by a series of transitions comprising at least one radiative transition involving emission of a photon of wavelength in the visible part of the spectrum.

Luminescence of most inorganic solids involves impurities, e.g., dopants, or structural defects. If the impurity or defect is the radiation-emitting entity, it is referred to as an "activator," and we will follow this usage herein. The presence of a second species of impurity or defect in the activator-containing host material often affects the emission characteristics of the material. If the second species absorbs energy from the exciting means and transfers part of the energy to the activator, with the reverse energy transfer being small, then the luminescent efficiency of the material is typically enhanced. In such a case, the second species is generally referred to as a

"sensitizer," and we will also follow this usage herein.

Although single crystal phosphors tend to have luminescent properties whose gross features are similar to those of powder phosphors of the same composition, the detailed features tend to differ in a generally unpredictable manner. Several reasons exist for this. For instance, since a powder phosphor is typically formed by a different process than the single crystal phosphor, chemical differences (e.g., different phases) may exist. Also, the crystal lattice in powder particles can be expected to be heavily strained, with a high defect density, whereas the lattice of a single crystal phosphor typically is relatively strain free and free of defects. Since luminescence is quite sensitive to the details of the crystal field, these lattice differences can lead to significant differences in the luminescence.

One of the materials whose luminescent properties have been investigated extensively is $\text{Y}_3\text{Al}_5\text{O}_{12}$, yttrium aluminum garnet (YAG). In particular, rare earth doped (including Ce^{3+} as well as Tb^{3+} doped) YAG has been investigated. (See, for instance, D. J. Robbins et al, Physical Review B, Vol. 19(2), pp. 1254-1269, (1979).) The energy transfer from different sensitizer species to a variety of activator species was also studied in YAG. In particular, the transfer from Ce^{3+} to Tb^{3+} in YAG powder has been studied (G. Blasse and A. Brill, The Journal of Chemical Physics, Vol. 47(6), pp. 1920-1926, (1967)). Also, YAG phosphor doped with Ce and Tb was prepared in powder form by calcination. Chemical Abstracts, Vol. 81, 97658d, page 421 (1974) and *ibid.*, 97659e. Japanese Patent 50-97590, Y. Fukuda et al, disclosure date August 2, 1975, also discloses powder YAG phosphor doped with Ce and Tb, and teaches that addition of Tb to Ce-containing YAG resulted in increased brightness. In particular, it indicates that the brightness improvement is due to the additional emission from Tb, and that the emission time of

Ce is not increased due to the presence of Tb.

Epitaxially grown monocrystalline Ce- or Tb-doped YAG CRT phosphor screens were also investigated (e.g., J. M. Robertson and M. W. van Tol, Applied Physics Letters, Vol. 37(5), pp. 471-472, (1980)). Saturation effects of the CL in such layers were also determined (e.g., W. F. van der Weg and M. W. van Tol, Applied Physics Letters, Vol. 38(9), pp. 705-707 (1981)).

Because of the great potential of high brightness visual displays, a phosphor with broadband emission that has high conversion efficiency, high power capability, high quench temperature, permits high resolution, and is not subject to substantial degradation is of considerable technological significance. This application discloses such a phosphor.

Definitions

"Yttrium aluminum garnet" or "YAG" is intended herein to include garnet material in which Sc, Ga, or In is substituted for part of Al, and one or more 4f-type rare earths, other than Ce or Tb, are substituted for part of Y. It is also intended to include such material in which equal parts of II/IV elements (e.g., Ca/Si, or Ca/Ge) are substituted for part of Y and Al, respectively.

The "intensity" of luminescent emission (in a given spectral range) from a given species, e.g., Ce, in a phosphor herein is proportional to the total number of photons emitted (in the spectral range) from the phosphor per unit time, under given excitation conditions, typically DC e-beam excitation. If two or more species emit in the spectral range, then the term "intensity of emission from X" or its equivalents implies that the observed total intensity was corrected to yield the intensity of emission from species X, whereas the sum of the various emissions will be referred to as the "total intensity of emission."

The "emission time" of an emitting species X herein is the time, measured from termination of the external excitation, required for the intensity of emission

from X to decrease to 5% of the maximum value.

Summary of the Invention

The inventive visual display system comprises an excitation beam source (electromagnetic radiation or electron beam), beam deflection means, beam modulation means, and a screen comprising single crystal phosphor material with broad band emission. The inventive phosphor comprises single crystal garnet material comprising Tb^{3+} as sensitizer and Ce^{3+} as activator, the garnet material being substantially of composition

$\text{Y}_{3-x-y-z}\text{Ce}_x\text{Tb}_y\text{RE}_z\text{Al}_{5-w}\text{X}_w\text{O}_{12}$, with $0 < x \leq 0.06$, $0 < y \leq 0.5$, $0 \leq z \leq 2$, and $0 \leq w \leq 3$, with RE being one or more rare earths (4f-type) other than Y, Ce, and Tb, and X being Sc, Ga, or In. Inventive phosphors can also comprise pairs, in equal parts, of a column II and a column IV element (e.g., Ca and Ge or Si). We will not specifically indicate the presence of the II/IV elements.

We have found that the broadband (Ce) emission from single crystal phosphors of the indicated composition has unexpectedly high intensity and long decay time. Typically, the Ce emission intensity in the 480-700 nm spectral range is at least 10% greater, and the Ce emission time at least 10% longer, in phosphors according to the invention than in analogous (i.e., same x , y , and w) Tb-free YAG phosphor identically excited. The observed increase in emission time implies that energy transfer from Tb to Ce occurs in phosphors according to the invention.

In typical embodiments the excitation beam is an electron beam, the screen comprises a single crystal [111] oriented YAG substrate with epitaxial phosphor layer thereon. Preferred phosphor compositions are

$\text{Y}_{3-x-y}\text{Ce}_x\text{Tb}_y\text{Al}_5\text{O}_{12}$, with $0.03 \leq y \leq 0.4$, and $0.005 \leq x \leq 0.03$, and

$\text{Y}_{3-x-y-z}\text{Ce}_x\text{Tb}_y\text{Lu}_z\text{Al}_{5-w}\text{Ga}_w\text{O}_{12}$, with x and y having the same

ranges as above, $0.05 \leq w \leq 3$, and $0.2 \leq z \leq 2$. We find that often good crystalline quality on YAG substrates results if z is about 1.6 times w . This ratio will, however, typically be different if a different substrate is used.

5 On a YAG substrate, the inventive phosphor is advantageously grown by LPE. In this case, melt temperatures between about 1030°C and about 1080°C, with growth rates of about 2-6 $\mu\text{m}/\text{min}$, have been found to yield useful results.

10 Brief Description of the Drawings

FIG. 1 shows the CL spectrum of single crystal Ce:YAG;

FIG. 2 shows the CL spectrum of single crystal Tb:YAG;

15 FIG. 3 gives the CL emission spectra of an exemplary single crystal Ce:Tb:YAG phosphor, and, for comparison, of Ce:YAG;

FIG. 4 shows the energy level diagrams for Ce and Tb in YAG;

20 FIG. 5 gives the CL emission spectra of single crystal Ce:Tb:YAG containing Lu and Ga, and, for comparison, of Ce:YAG containing Lu and Ga;

FIG. 6 shows the time resolved Ce emission intensity for exemplary single crystal Ce:Tb:YAG and Ce:YAG phosphors;

25 FIG. 7 presents the time-resolved Tb emission intensity for exemplary single crystal Lu- and Ga-containing Ce:Tb:YAG and Tb:YAG;

FIG. 8 shows the Ce-emission intensity of Ce:Tb:YAG as a function of Tb content; and

FIG. 9 schematically depicts apparatus for the practice of the invention.

Detailed Description

35 A central aspect of the invention resides in the novel phosphor, namely, single crystal YAG doped with both cerium and terbium, to be referred to as single crystal Ce:Tb:YAG.

We have made the unexpected discovery that in such material significant energy transfer from Tb to Ce occurs, in contrast to the powder Ce:Tb:YAG case. This transfer results in substantially increased intensity of Ce emission, as compared to single crystal Ce:YAG. Tb thus acts as sensitizer in the inventive phosphor, with Ce being the activator. The Tb-sensitization manifests itself also in increased Ce emission time, as compared to prior art single crystal Ce:YAG and powder Ce:Tb:YAG.

FIG. 1 shows the CL spectrum from a (prior art) single crystal sample of Ce:YAG. The Figure illustrates the broadband-nature of Ce emission in the 480-700 nm spectral region. Similarly, FIG. 2 shows a (prior art) CL spectrum from single crystal Tb:YAG, demonstrating that Tb has essentially a line emission spectrum in the same spectral region. It is to be noted that $\text{Y}_3\text{Al}_5\text{O}_{12}$ does not show significant luminescence in the above spectral region.

FIG. 3 shows an exemplary CL spectrum 31 of single crystal Ce:Tb:YAG, of approximate composition $\text{Y}_{2.975}\text{Ce}_{0.008}\text{Tb}_{0.017}\text{Al}_5\text{O}_{12}$, and, for comparison, the spectrum 30 of single crystal Ce:YAG, of approximate composition $\text{Y}_{2.99}\text{Ce}_{0.01}\text{Al}_5\text{O}_{12}$. Both spectra were obtained under identical conditions, and are plotted to the same scale. It is evident that spectrum 31 not only contains Tb emission lines but that the broadband Ce emission is substantially increased over spectrum 30. FIG. 3 thus exemplifies the increased Ce emission intensity of single crystal Ce:Tb:YAG, as compared to Ce:YAG. The Ce emission chromaticity of the total emission from the Ce:Tb:YAG sample corresponded to $x=0.375$, $y=0.528$ on the well-known CIE x,y Chromaticity Diagram (see, for instance, R. S. Hunter, The Measurement of Appearance, pp. 81-97, John Wiley, New York (1975)).

FIG. 4 gives simplified energy level diagrams for Ce and Tb in YAG. A Tb^{+3} ion, excited into the $^5\text{D}_3$ state, can lose energy by direct transition to a ^7F or the $^5\text{D}_4$ level, or, under appropriate conditions, to transfer energy to a Ce^{+3} ion, raising it to the $^2\text{D}_{3/2}$ level. Similarly, a Tb^{+3} ion in $^5\text{D}_4$ can decay to a ^7F state, or, under the right conditions, to raise a Ce^{+3} ion to $^2\text{D}_{3/2}$. The notation used is standard and well known to those skilled in the art. Herein we often do not specifically note the valence state of Ce and/or Tb, but the +3 valence state is intended to be implied.

FIG. 5 shows the spectral distribution 51 of the Ce emission from a phosphor according to the invention, namely, a single crystal Lu- and Ga-containing Ce:Tb:YAG of approximate composition

$\text{Y}_{1.958}\text{Lu}_{0.93}\text{Tb}_{0.10}\text{Ce}_{0.012}\text{Al}_{4.24}\text{Ga}_{0.76}\text{O}_{12}$. For comparison, the spectrum 50 of single crystal

$\text{Y}_{2.058}\text{Lu}_{0.93}\text{Ce}_{0.012}\text{Al}_{4.24}\text{Ga}_{0.76}\text{O}_{12}$ is also shown, demonstrating the enhanced broadband emission from the Tb-containing sample. The CIE chromaticity coordinates of the emission from the latter are about $x=0.322$, $y=0.578$.

Incorporation of Tb does not only result in increased Ce emission but also in an increase in the Ce emission time, as is exemplified by the experimental results shown in FIG. 6. Curves 60 and 61 represent the normalized light output at 530 nm of single crystal

$\text{Y}_{2.987}\text{Ce}_{0.013}\text{Al}_5\text{O}_{12}$ and $\text{Y}_{2.87}\text{Ce}_{0.013}\text{Tb}_{0.117}\text{Al}_5\text{O}_{12}$, respectively, as a function of time, following excitation by means of a 2 μsec 20 KV electron pulse, with $t=0$ corresponding to the termination of the excitation. The emission times $\tau_e^{(60)}$ and $\tau_e^{(61)}$, i.e., the times at which the light output has decayed to 5% of the maximum, are about 8 μsec and 12.5 μsec , for curves 60 and 61, respectively. In addition to having substantially

increased emission time, the Tb-containing sample has a containing sample has a considerably enhanced emission tail, as compared to the Tb-free sample. Both curves 60 and 61 start at about 40% at $t=0$. The two curves have essentially identical behavior in their increasing parts as well as in the initial portion of the decay part. These parts of the curves are not shown in the Figure. Increasing and decreasing portions of the curves are identified by means of arrows adjacent to the curves.

FIG. 7 shows exemplary data demonstrating that the energy transfer from Tb to Ce results in considerably reduced Tb emission and somewhat decreased Tb emission time. Curves 70 and 71 represent the Tb emission at 544 nm single crystal $Y_{1.97}Lu_{0.93}Tb_{0.10}Al_{4.24}Ga_{0.76}O_{12}$ and $Y_{1.958}Lu_{0.93}Tb_{0.10}Ce_{0.012}Al_{4.24}Ga_{0.76}O_{12}$, respectively. The curves are plotted to the same scale, $t=0$ corresponding to the termination of the excitation, a 2 μ sec 20 KV electron pulse.

An exemplary experimental curve of Ce emission intensity as a function of Tb content of Ce:Tb:YAG is shown in FIG. 8, indicating that the presence of a relatively small amount of Tb, e.g., corresponding to about $y=0.0025$, can produce a significant increase (typically about 10%) in the Ce emission intensity in single crystal Ce-containing YAG.

The observed increase in Ce emission from single crystal Ce:Tb:YAG is not suggested by the prior art. For instance, Japanese Patent 50-97590 (op. cit.), discloses that in powder Ce:Tb:YAG the observed improvement in brightness over Ce:YAG is due to "afterglow" of Tb, with the Ce emission time in powder Ce:Tb:YAG being equal to or less than that in Ce:YAG. Since Tb has a much longer emission time in YAG than Ce, the fact that in powder Ce:Tb:YAG the Ce emission time was observed to be

essentially the same as in Ce:YAG shows that no significant Tb to Ce energy transfer occurred in the prior art powder Ce:Tb:YAG.

5 The fact that in single crystal Ce:Tb:YAG terbium is an efficient sensitizer, is of considerable practical importance. Although there exist applications of phosphors in which a line spectrum such as that of Tb:YAG is useful or even required, e.g., in holographic displays, there are other display applications, (often referred to as
10 "refractive" displays), e.g., color displays using frame sequential filtering, or refractive displays that use different parts of a broadband spectrum to display different information, which depend on the availability of a high brightness phosphor with a broadband emission
15 spectrum. It is in such applications that the inventive phosphor is most advantageously employed.

Doping of Ce:Tb:YAG with one or more rare earths can result in a shift of the chromaticity of the emission. In particular, this is the case for Lu-doping, as
20 exemplified by the different CIE coordinates of the samples of FIGS. 3 and 5. The color of Lu-containing Ce:Tb:YAG makes such material of interest in, e.g., color projection TV or for video screens.

Apparatus for the practice of the invention is
25 schematically depicted in FIG. 9, wherein an excitation beam 91 is emitted by beam source 90, is amplitude modulated by modulator 92, the modulated beam 93 is appropriately deflected by deflector 94, and impinges on screen 95, with luminescent radiation 96 emitted from the
30 excited region of the screen. The beam can be either an electron beam or electromagnetic radiation, typically in the UV.

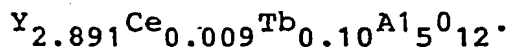
Although the use of any method capable of producing single crystal Ce:Tb:YAG is contemplated by us, a
35 convenient method, and indeed the one currently considered most suitable, is growth of Ce:Tb:YAG on YAG by liquid phase epitaxy (LPE). LPE has reached a high state of

development in, e.g., the growth of magnetic bubble materials, and those skilled in the art of crystal growth are generally familiar with the technique. See, for instance, S. L. Blank and J. W. Nielsen, Journal of Crystal Growth, Vol. 17, pp. 302-311, (1972), and J. W. Nielsen, IEEE Transactions on Magnetics, MAG-12, pp. 327-345, (1976).

LPE has been applied to the growth of single crystal phosphor layers, including Ce:YAG and also Tb:YAG (J. M. Robertson et al, Philips Journal of Research, Vol. 35, pp. 354-371, (1980)). As FIG. 2 of J. M. Robertson and M. W. van Tol, Applied Physics Letters, Vol. 37(5), pp. 471-472, (1980), teaches, single crystal Ce:YAG of [111] orientation has maximum light output when grown at about 1010°C, and the light output of single crystal [111] Tb:YAG is approximately flat for growth temperatures between about 1010°C and about 1040°C, with a moderate maximum for a growth temperature of at about 1070°C. These teachings suggest that [111] Ce:Tb:YAG, the preferred material, might be most advantageously grown at about 1010°C, since this growth temperature could be expected to maximize light output in the Ce band. However, we found that the highest emission efficiency of at least some inventive phosphors resulted for growth temperatures between about 1030°C and about 1080°C. For typical materials we find that the emission efficiency of Ce:Tb:YAG with $z=w=0$ has a peak for growth at about 1065°C, and drops by about 20% at the limits of the above range. For material containing Lu and Ga the growth temperatures are typically somewhat lower.

Example 1: A melt consisting of Y_2O_3 , CeO_2 , $Tb_{407}Al_{203}$, PbO , and B_2O_3 (9.200, 3.956, 0.374, 15.592, 1400, and 28.30 gm, respectively) was prepared in a platinum crucible in air at ambient pressure. The starting materials were commercial grade, of typically 99.999% purity. Standard LPE growth (melt temperature 1066°C, growth rate 5.36 $\mu\text{m}/\text{min}$, with substrate

rotation at 60 rpm, rotation reversal after each revolution) produced a 6.44 μm thick epitaxial single crystal Ce:Tb:YAG layer on a 3.8 cm (1.5 inch) diameter [111] YAG wafer. The composition of the layer was approximately



The sample was mounted in a demountable CRT and exposed to a focussed beam of 20 KV electrons. The luminescent emission was analyzed with standard equipment, including a Jarell-Ash 0.5 m spectrometer, and a Hamamatsu R955 photomultiplier cooled thermoelectrically to -10°C . The emitted radiation had a spectral distribution similar to that shown in FIG. 3. The intensity of Ce emission in the 480-700 nm spectral range was about 1.5 times that of a similarly prepared single crystal Ce:YAG sample (approximate composition $\text{Y}_{2.991}\text{Ce}_{0.009}\text{Al}_{50}\text{O}_{12}$). By means of pulsed excitation (2 μsec pulses of 20 KV electrons, 100Hz repetition rate) the Ce emission time is found to be about 12 μsec , which is about 1.5 times the Ce emission time observed in Ce:YAG.

Example 2: A single crystal layer of composition $\text{Y}_{1.958}\text{Lu}_{0.93}\text{Tb}_{0.10}\text{Ce}_{0.012}\text{Al}_{4.24}\text{Ga}_{0.76}\text{O}_{12}$ was grown by LPE on [111] YAG. The melt composition was $\text{Y}_2\text{O}_3 = 3.504$, $\text{CeO}_2 = 3.621$, $\text{Tb}_4\text{O}_7 = 0.885$, $\text{Lu}_2\text{O}_3 = 9.630$, $\text{Al}_2\text{O}_3 = 8.488$, $\text{Ga}_2\text{O}_3 = 13.161$, $\text{PbO} = 1200$, and $\text{B}_2\text{O}_3 = 30.81$ gm. Growth temperature was 1029°C , growth rate $5.77 \mu\text{m}/\text{min}$, and layer thickness $8.65 \mu\text{m}$. Growth and luminescence measurement was substantially as described in Example 1. The emission spectrum was similar to the spectrum shown in FIG. 5. The Ce emission intensity was 1.48 times the Ce emission intensity of a single crystal phosphor of composition

0142931

- 13 -

$\text{Y}_{2.058}\text{Lu}_{0.93}\text{Ce}_{0.012}\text{Al}_{4.24}\text{Ga}_{0.76}\text{O}_{12}$. The emission time exceeds that of the Tb-free material by at least 10%.

5

10

15

20

25

30

35

Claims

1. Visual display apparatus comprising
an excitation beam source, the beam selected from
electron beams and beams of
5 electromagnetic radiation comprising UV radiation,
a screen comprising single crystal phosphor
material, the beam caused to be incident on the screen,
and
means for deflecting the beam, and means for
10 amplitude modulating the beam,
CHARACTERIZED IN THAT
the phosphor material comprises single crystal
material substantially having the composition
 $Y_{3-x-y-z}Ce_xTb_yRE_zAl_{5-w}X_wO_{12}$, with RE
15 being one or more rare earths of the 4f-type, other than Y,
Ce, and Tb, X is selected from Sc,
Ga, and In, $0 < x \leq 0.06$, $0 < y \leq 0.5$, $0 \leq z \leq 2$, and $0 \leq w \leq 3$.
2. Apparatus according to claim 1,
CHARACTERIZED IN THAT
20 the phosphor material is an epitaxial layer formed
by liquid phase epitaxy on a single crystal substrate.
3. Apparatus according to claim 2,
CHARACTERIZED IN THAT
the substrate material is substantially of
25 composition $Y_3Al_5O_{12}$.
4. Apparatus according to claim 3,
CHARACTERIZED IN THAT
the substrate substantially has [111] crystalline
orientation.
- 30 5. Apparatus according to any one of preceding
claims 1-4,
CHARACTERIZED IN THAT
RE is Lu and X is Ga.
- 35 6. Apparatus according to claim 5,
CHARACTERIZED IN THAT
 $0.005 \leq x \leq 0.03$, $0.03 \leq y \leq 0.4$, $0.2 \leq z \leq 2$, and $0.5 \leq w \leq 3$.

7. Apparatus according to claim 6,

CHARACTERIZED IN THAT

the substrate is substantially of composition

$\text{Y}_3\text{Al}_5\text{O}_{12}$ and z is about $1.6w$.

5 8. Apparatus according to any one of preceding claims 1-4,

CHARACTERIZED IN THAT

$0.005 \leq x \leq 0.03$, $0.03 \leq y \leq 0.4$, $z=0$, and $w=0$.

10 9. Apparatus according to any one of preceding claims 1-8,

CHARACTERIZED IN THAT

the beam consists substantially of ultraviolet radiation.

15 10. Apparatus according to any one of preceding claims 1-9,

CHARACTERIZED BY

means for projecting at least a part of the radiation in the 480-700 nm spectral range emitted by the phosphor onto a display screen.

1/4

FIG. 1
(PRIOR ART)
Ce: YAG

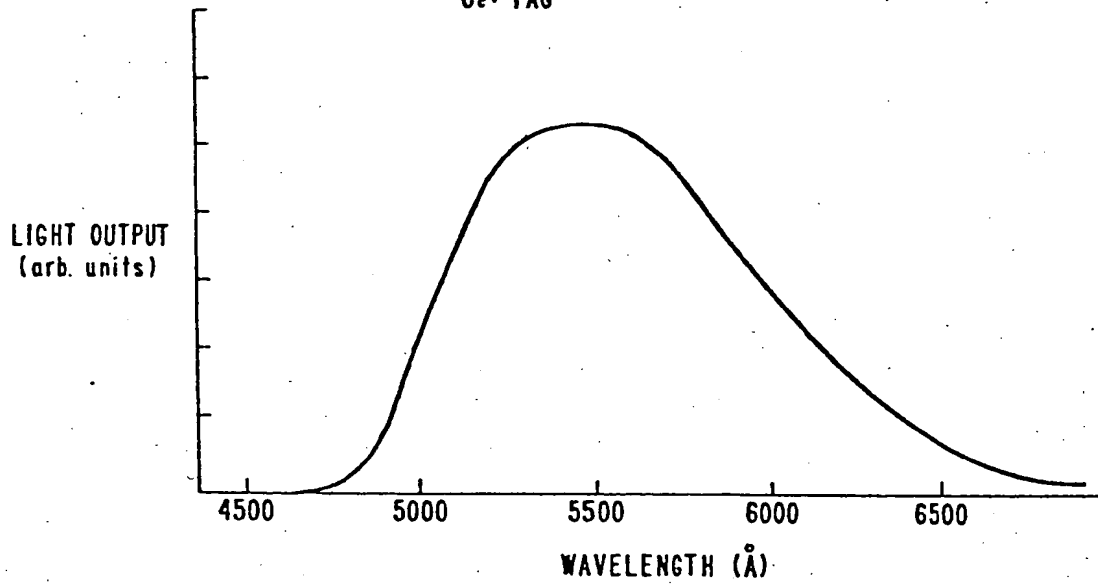
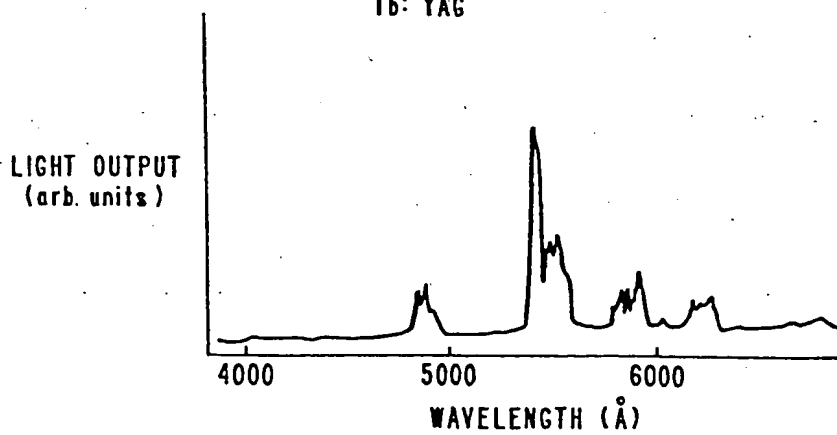


FIG. 2
(PRIOR ART)
Tb: YAG



2/4

FIG. 3

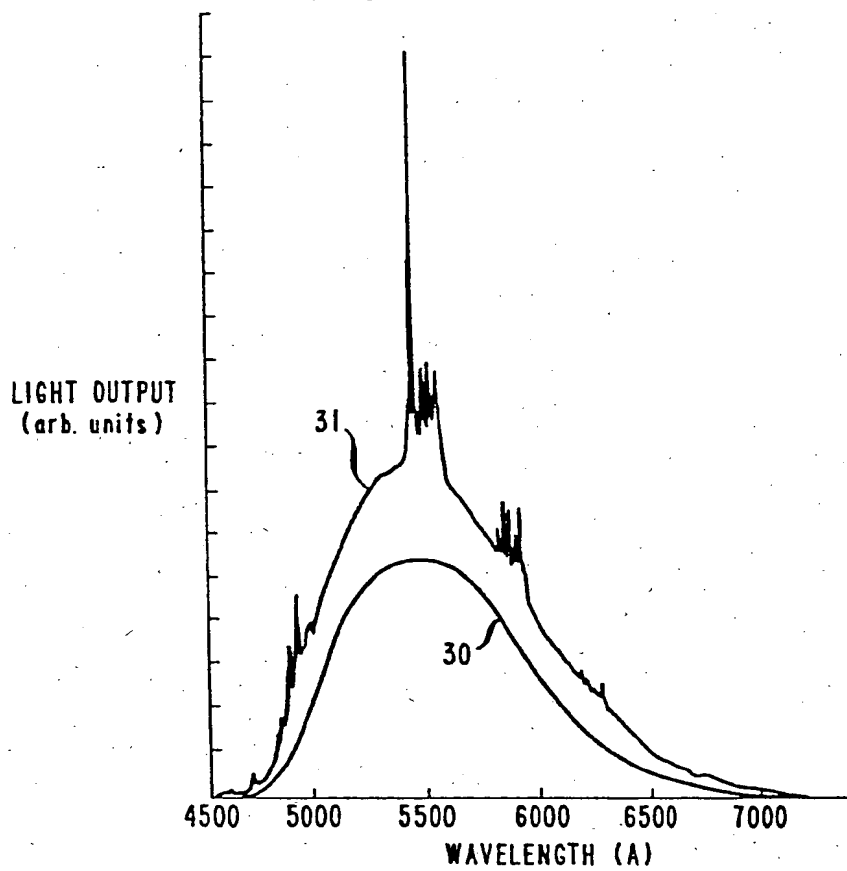


FIG. 4

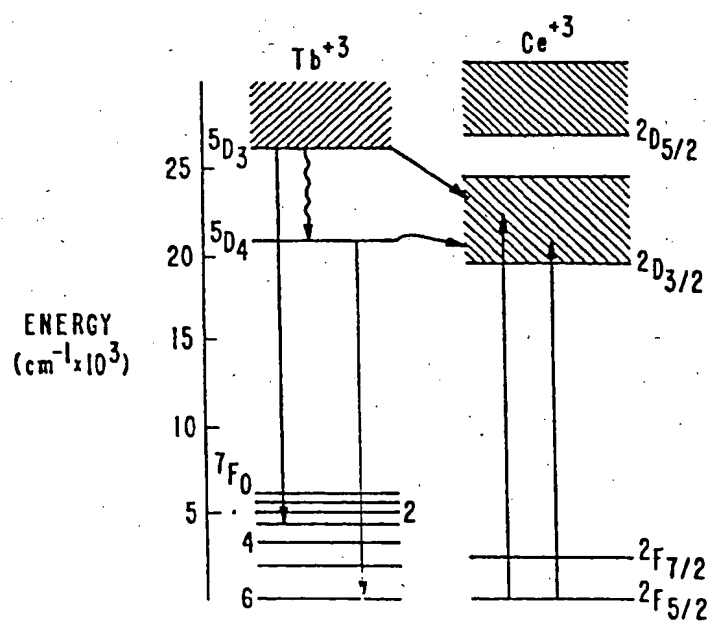


FIG. 6

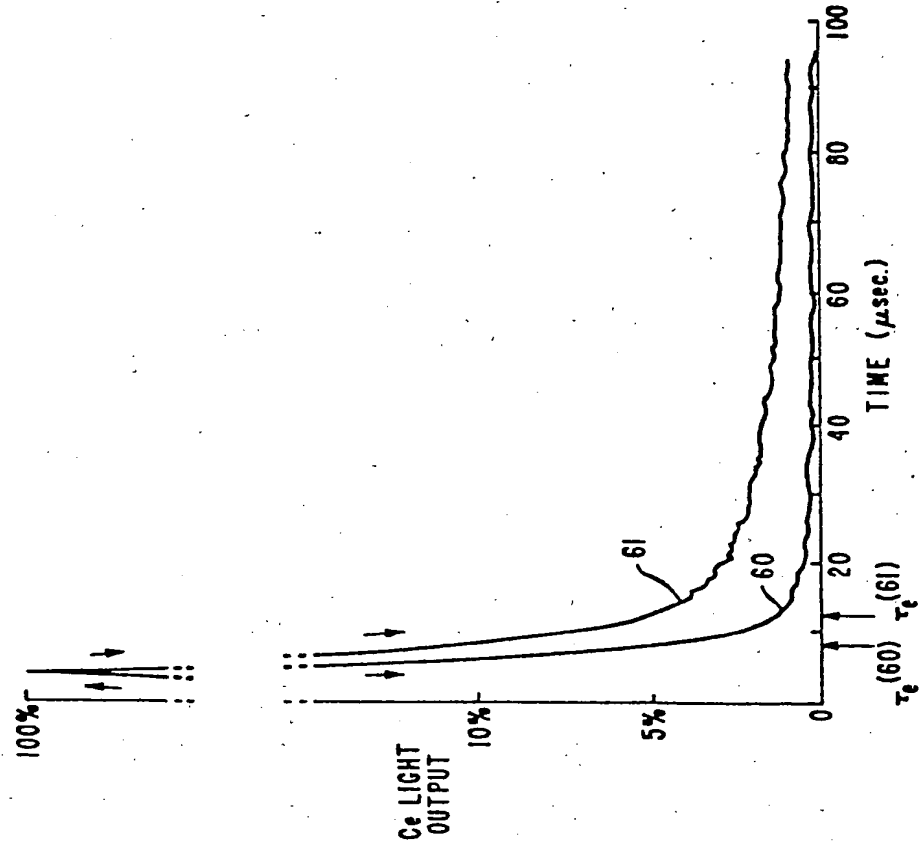
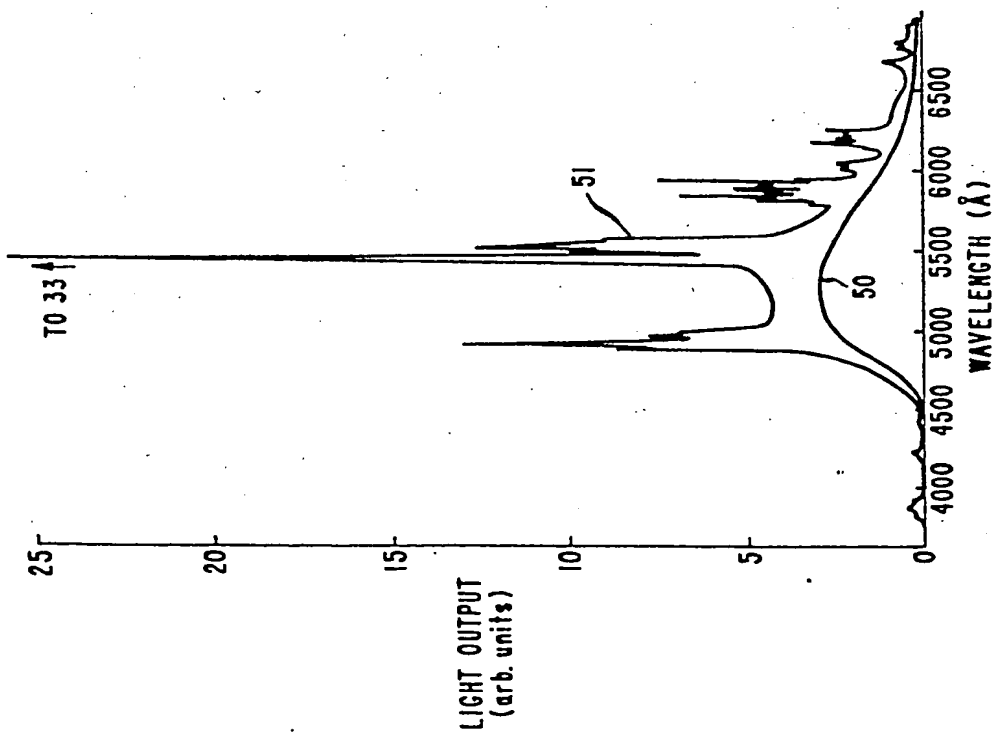


FIG. 5



4/4

FIG. 7

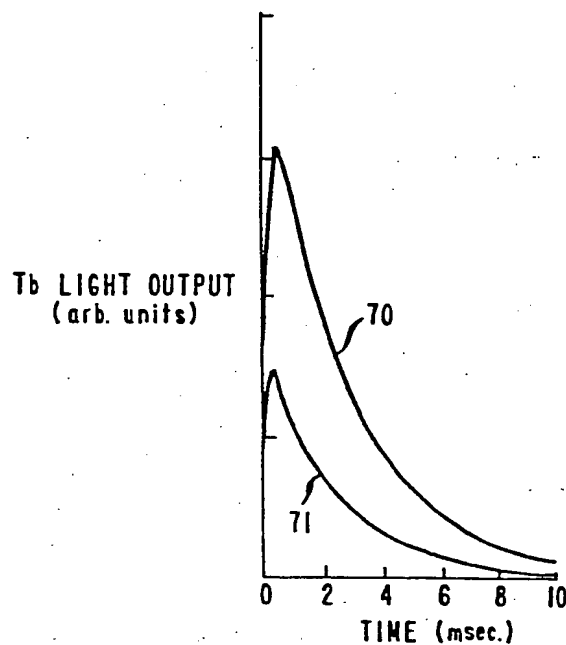


FIG. 8

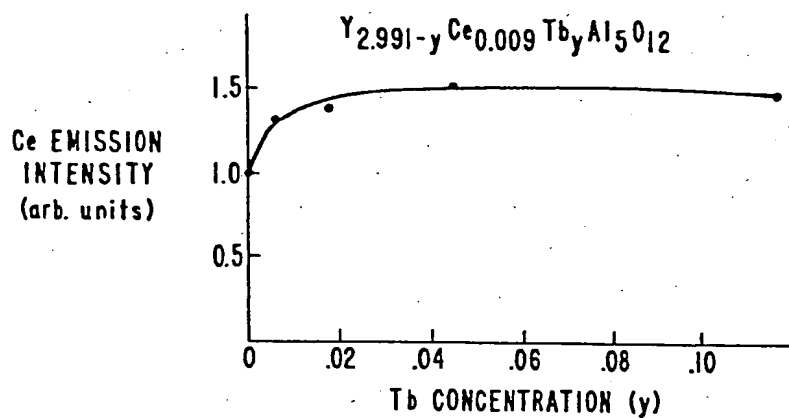
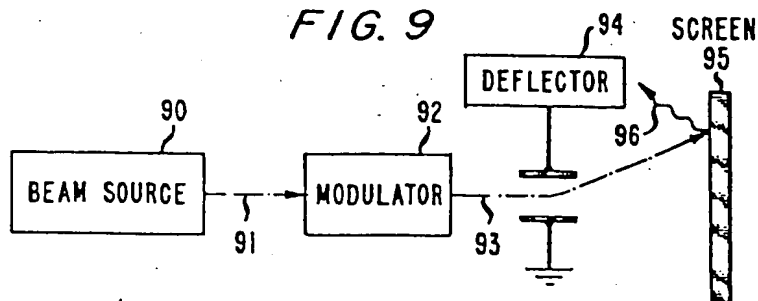


FIG. 9





European Patent
Office

EUROPEAN SEARCH REPORT

0142931

Application number

EP 84 30 6854

DOCUMENTS CONSIDERED TO BE RELEVANT			
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl.4)
A	DE-A-2 739 437 (J. MATTHEY & CO.) * claims 1-10 *	1	C 09 K 11/475
A	US-A-4 180 477 (R. BARNES) * claims 1,2; column 2, lines 1-13 *	1	
			TECHNICAL FIELDS SEARCHED (Int. Cl.4)
			C 09 K 11/00
The present search report has been drawn up for all claims			
Place of search THE HAGUE		Date of completion of the search 14-12-1984	Examiner DROUOT M.C.
CATEGORY OF CITED DOCUMENTS			
X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document		T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons & : member of the same patent family, corresponding document	